DETERMINATION OF VARIATIONS IN CERTAIN MECHANICAL PROPERTIES ACROSS THE WIDTH OF A FINISHED PRODUCT OF POLYPROPYLENE WEB

Ву

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CHAPTER I

INTRODUCTION

A web is a long continuous film of material with its length much greater than its width or thickness. Used in most high production rate plants, webs can be any material, from paper to steel, that can be processed on a continuous basis. This rapid handling of material benefits the user by reducing time of production, but the web form can sometimes cause serious problems.

Out of plane deformation of the web, e.g. waves, can ruin the finished product, costing the producer untold dollars in not only time but material. If some way can be determined to predict when waves occur, then methods of prevention could be investigated. Some base information has to be established before one can begin to predict such a complicated instability.

The aspects of the web that should be investigated include: yield strength, Young's modulus and anistropy. Usually, averages of these values are reported, but local values are also needed to determine how processing is affecting the web's properties. A profile of these properties across the width of the final product could provide useful information on the local attributes that may

lead to waves and other instabilities.

It is proposed that the processing may cause variations across the web and that under certain conditions in handling or winding, these may be extremely harmful to the final product. This probe determines what variations in mechanical properties across the width of a finished product of web exist and develops techniques to measure as many of these variations in a static testing mode as possible to help prevent disastrous results. Once the techniques have been established and data is obtained, methods of presenting the information in a useful manner will be explored. Anistropy, local mechanical properties, and other parameters will be investigated to ascertain their effects, if any, on the onset of wave formation in association with the biaxial orientation process.

CHAPTER II

BACKGROUND

Material properties of polypropylene and polyester vary with respect to the history of their processing. Young's moduli of polypropylene range from 100 to 575 ksi [19]. The Young's moduli for polyester range from 280 to 3,500 ksi depending on the type of reinforcement. The maximum yield strengths are 4.5 to 5.4 ksi and 23 ksi for polypropylene and polyester, respectively [19].

"The glass transition is associated with the amorphous phase of a polymer and occurs at a temperature Tg at which segments of the main chain become mobile" [2]. Polypropylene is brittle below this temperature before the biaxial orientation is performed. The range of glass transition temperature for polypropylene is -30 C to 20 C, depending on tacticity and thermal history [2]. Polyester's glass transition ranges from 73 C to 81 C [19].

Biaxial orientation is the process that produces equal or balanced material properties in two perpendicular directions [21]. In polymers, this is achieved by heating the material above the glass transition temperature (to allow greater ease of chain disentanglement and slippage) while stretching in the MD until the desired MD modulus and

yield strength are obtained. Then stretching the material in the CD is utilized until the desired MD and CD properties are equal. This stretching process is akin to work hardening, also known as strain hardening in metals. As the polymer is stretched, higher values for the properties are achieved (see figure 2.1). "Strain hardening continues by molecular alignment (sometimes called cold drawing) up to the fracture stress" [22]. From a random orientation, the polymeric chains are aligned approximately parallel to the applied stress [8]. This alignment brings the chains closer together where mutual attraction increases, especially when the chains are polar or symmetrical. As the stretching takes place, bond stretching or valence angle deformation also results. This is the elastic component of the biaxial stretching and is recovered when the applied stress is removed. Bond stretching, the viscous flow of molecular chains past one another, causes friction losses that cannot be recovered. All three molecular activities must be considered when determining the amount of overall elongation needed for the desired product. The main goals of the biaxial orientation process is to achieve molecular alignment of polymeric chains which increases strength, decreases brittleness and eliminates anistropy.

When dealing with polymers, viscoelastic effects must be examined. Viscoelasticity characteristics are between those of metals and viscous fluids. The models shown in





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figure 2.2 represent viscoelasticity. Keep in mind that this model is a highly simplified representation of the very complex polymer response. The springs, each with a different spring constant, in the models symbolize the elastic portions of the response which include bond stretching, crystallinity and chain rigidity [19]. Representing the fluid time dependent response, the dashpots each have a different viscosity. Figure 2.3 shows the standard time versus strain viscoelastic response. Stress and strain in this type of a model are time, rate of loading and temperature dependent. "Plastics have no true proportion limits" [19]. Most polymers have a curvilinear stress-strain curve. Depending on the polymer, test temperature and thermal history, length of the linear segment of the curve can vary greatly. Using moduli obtained from these linear regions, the prediction of deformation is very difficult. A polymer's yield point does not indicate the onset of "significant flow" as it does for most metals. The yield point is defined, in the case of polymers, in the general area of the "knee" in the engineering stress-strain curve. This "knee" is due to thinning of the cross-sectional area of the polymer in the tensile test. If true stress, based on instantaneous cross-sectional area rather than initial area, were plotted versus true strain, the "knee" would be much less defined. It is interesting to note that in general polymers' Poisson's ratios are much greater than those found in



Figure 2.3 Creep response of four element model. [22]

metals resulting in much greater necking in polymers.

Figure 2.4 represents the Voigt model. At t=0, there is a step jump in stress due to the dashpot in figure 2.2 which appears to have infinite stiffness. As time progress from 0 to 100 seconds in figure 2.3, stress is held constant while strain continues to increase. This is called creep. "Creep curves generally show three continuous stages: a first stage marked by large and rapid initial deformation; a second stage where deformation continues at a relatively slow, constant rate; a third stage in which rupture occurs" [19]. At t=100 seconds, the strain is held contant while the stress drops in a stepwise fashion. Stress relaxation then ensues as strain is held constant. Stress relaxation can be approximated by:

$O(t) = O_0 e^{-t/T}$

...(1),

where σ_{\bullet} is the initial stress before stretching is stopped, t is time and T is relaxation time defined by π /E. π is the viscosity and E is Young's modulus.

Detection of local thinning can be done with the use of circles printed on the surface of a deforming specimen. Figure 2.5 demostrates how changes in the circle as it becomes an ellipse can give information about thinning [21]. The major axis indicates the major direction and magnitude of stretching. The engineering strain, in the direction of the major axis, can be measured by noting the length of the major axis of the ellipse and comparing it to



Figure 2.4 Voigt model's stressstrain diagram. [22]



Figure 2.5 Ellipse deformation [21]

the major axis of the original circle. The same can be done for the minor axis and its direction, but where the major strain is always positive for the case of a tensile test, the "minor" strain may be positive or negative. In the tensile test, the minor strain is induced by the Poisson effect and the constancy of volume in plastic flow of a noncompressible material. If there is an area change from the circle to the ellipse then the thickness has changed. For example, if the area of the ellipse increases, the thickness must decrease to maintain the constant volume of the continuum. "This can be easily observed by experimenting with a wide rubber band" [12]. Notice when stretching the rubber band, an ellipse forms and the thickness decreases. If the continuum is to exist then constancy of volume must prevail.

Several methods of testing are available to obtain the needed mechanical property values. One method is optical. Using laser speckle interferometry, the in-plane elastic constants can be obtained [15]. The out-of-plane elastic constants are determined by conventional or holographic interferometry results and the Maxwell Relationships. Optical methods can only measure displacements and Poisson ratios. "Nevertheless, when combined with the necessary stress information (provided by other means), in-plane Young moduli and shear modulus can also be measured" [15]. Another method of measuring elastic constants is by acoustical means. Using ultrasonic waves, the constants

can be determined by measuring the effects of the sound waves as they pass through the medimum in certain directions [12], [16] and [18]. This method can be used in dynamic testing, but the elastic constants are always higher than those produced by mechanical means. This is possibly due to the time involved in the two methods. In ultrasonic testing, there is very little time for viscoelastic relaxation to take place. Mechanical testing gives the material much more time for viscoelastic relaxation to occur. Mechanical testing, the last method examined for possible use, is accomplished by measuring extension, load applied and physical dimensions, from which a stress-strain curve is generated. The elastic contants and yield point can then be derived.

Waves are out-of-plane deformations. Several theories on their formation vary greatly. One theory is that the waves form similar to those in elastic columns in compression. Euler's formula [25] :

 $F_{cr} = \pi^2 E I / L^2 \qquad \dots (2)$

E is Young's modulus, L is the length of the column (the width in the case of films) and I is the moment of inertia of the cross section about the bending axis. Notice the critical force depends on E and geometry. In the case of thin films, a "critical" strain is caused by Poisson's effect. The tensile load applied to films causes a compression strain in the transverse direction that induces waves. It is possible that the strain that causes the

waves is somehow related to the E, L, and I of equation (2).

The Von Mises stress defines the start of yielding. Von Mises stress is represented by

 $\sigma' = (\Gamma(\sigma_1 - \sigma_2)^2 + (\sigma_2 - \sigma_3)^2 + (\sigma_3 - \sigma_1)^2]/2)^{1/2}$...(3) when a triaxial stress state exists where σ_1 , σ_2 and σ_3 are the principal stresses [26]. Two specific cases can exist: plane strain and plane stress. Plane strain is where strain exists in two principle directions but not in the third principle direction. In the case of plane stress, the stress exists in two directions and does not exist in the third. Thin films under tension in a web handling process can be under plane stress. The above equation reduces to

$$\sigma' = (\sigma_1^2 - \sigma_1 \sigma_2^2 + \sigma_2^2)^{1/2} \qquad \dots (4)$$

for plane stress. Plastic deformation occurs when the von Mises stress exceeds Y (uniaxial yield stress of the material). The following is the relationship equation:

$$\sigma^{\prime 2} = \chi^2 \qquad \dots (5).$$

Figure 2.6 shows a 3-D stress element with stresses in the x, y and z directions with corresponding shear stresses. This is a triaxial stress state. The plane stress case is illustrated in figure 2.6. Notice there is no stress in the z direction. Figure 2.7 shows how one of the principal stresses relates to stress and shear. Figure 2.8 shows how a Mohr's circle, where stress is plotted versus shear for



Figure 2.6 (a) 3-D Stress cube, (b) plane stress. [26]



Figure 2.7 Relationships of stresses in 3-D stress element to principal stress. [26]



Figure 2.8 Mohr's circle diagram. [26]



Figure 2.9 Two examples of stress (a) Plane stress, (b) Plane strain. [26]

the stress element, is constructed to yield the principal stresses. The principal stresses are those points of Mohr's circle where the stress is at its maximum and minimum. Plane strain is compared with plane stress in figure 2.9. Figure 2.10 shows Mohr's circles in various stress states.

Crazing can be an important part in the deformation of polymers. Yefimov, Bulayev, Ozerin, Rebrov, Godovskii and Bakeyev have done tests that show "crazing in polypropylene is accompanied by a considerable rise of polymer internal energy due to new surface created inside the crazes" [14]. They studied films of isotactic polypropylene 100 to 200 micrometers thick and 20 to 40 millimeters long at a relative elongation of up to 100 percent. The rate of deformation was 5 millimeters per minute. The samples were prepared by annealing oriented polypropylene at near its melting point. Using a microcalorimeter, they noticed heat was absorbed before crazes started forming. Small-angle X-ray scattering was used to detect crazing. A great number of crazes, 2.5 x 10⁵ per centimeter, formed when the stress reached the "limit of forced elasticity", σ_t . After craze formation began, a "substantial" amount of energy was generated. "The energy stored in PP elongated to 100% amounts to 20-25 J/g" [14]. They concluded the heat dissipated was much less than the deformation energy so that the net internal energy increased during crazing. They also state that the change



igure	2.10	Mohr's circles for various
-		states of stress.
		(a) Uniaxial tension.
		(b) Uniaxial compression.
		(c) Biaxial tension
		(plane stress).
		(d) Triaxial tension.
		(e) Biaxial compression
		with tension. [26]

in internal energy increases linearly with deformation.

Jenkins and Jenkins studied the induction of a prefered orientation in a liquid crystal co-polyester by extrusion and drawing [10]. They found that the "rod-like" molecules of co-polyester tend to align themselves along the fiber axis during hot stretching following extrusion. But what they discovered later has a great importance, the stiffness peaked when the heat treatment was 170 C. At temperatures greater than 170 C, the preferred orientation would decrease. In other words, the stiffness in the preferred orientation would drop. They concluded that the mechanical properties are critically dependent on thermal and strain history.

In this study, the tentering process takes place at elevated temperate as well. The tentering region may be effected by this phenomenon.

CHAPTER III

EXPERIMENTAL

All experimentation was performed using an Instron tensile testing machine in correlation with specially designed grips and a load cell-tranducer-plotter arrangement (figure 3.1). Set for a constant displacement rate of 0.5 inches/minute, the Instron extended the sample positioned between the two roller grips. Each grip consisted of a steel bar bounded by "channel iron" supports. Down the length of the steel bar, a thin straight groove was cut at a depth of one half of the diameter of the bar. This groove is used to prevent the sample from slipping from the grip. Through the "channel iron" support and the steel bar, holes were drilled and removable bolts fitted stopping the bar's rotation during testing and allowing easy sample mounting and removal. The load cell-transducer-plotter set-up recorded voltage, proportional to load, versus time once testing began. System calibration was three-fold: first, measuring actual displacement of the Instron versus time; second, affixing a known load to the system and noting plotter displacement; third, comparing real time against plotter pin movement (see appendix for calibration results).



Figure 3.1 Instron and associated equipment.

The heating of samples to a specific temperature and maintaining that temperature (figure 3.2) during testing was achieved by the use of strip heaters, a Plexiglas heat reflector and a PID (Proportional Integral Derivative) controller (see figure 3.3).

Specimens of both polypropylene and polyester were tested. Polypropylene was taken from a "frozen" production line (figure 3.4) and samples were secured from sections of the finished product. Sampling along the length of the transverse direction orientation (TDO) (figure 3.5), provided information on the degree orientation or anistropy occurring as the polypropylene is being elongated in the cross-machine direction (CD). At ten foot intervals from the exit of the TDO, samples were cut in both the machine direction (MD) and CD to give data concerning change in anistropy as the material process attempts to approach desired biaxial material orientation. Specimens were also taken in the finished product across the width of the web to form a profile of data concerning uniformity (figure 3.6 & 3.7). Some samples were taken at a 45-degree angle to the MD to test biaxiality. Not knowing the past processing history of polyester and its size limitations, tests were run only in the MD. Experiments were also done at elevated temperatures to observe changes in material properties at different temperatures (figure 3.8).

Preparation of samples followed this format. First, the specimens were cut with a new razor blade after the



Figure 3.2 Schematic of PID controller and heater set-up.



Figure 3.3 PID Controller.



Cont'd.









Figure 3.6 Across the web sampling between winder and treater.



Figure 3.7 Sampling between winder and treater.



Figure 3.8 Polyester specimen being tested at an elevated temperature.

appropriate section and direction of the material was marked with a permanent marker. Care was taken not to damage the integrity of the material's edge. Specimen size was typically 6 x 24 inches, although final size may be altered slightly to remove sharp edge stress concentrations. Second, the 6 inch ends are covered with 1/4 inch of masking tape promoting sample mounting into grip groove. Third, using a permanent marker and a circle template, 1 inch diameter circles are drawn onto the surface of the specimen, again being careful not to damage the material's surface.

The samples are mounted by inserting the taped ends into the bar's slot and then rolling the bar until the specimen has been wound around approximately three times. (Note: on thicker materials, greater than 2 mils, adhesive can be applied onto the bar and specimen before wrapping to ensure traction as long as the adhesive is much stiffer than the sample.) This is repeated for both top and bottom grips. When winding the bar, care is taken to leave the sample in an unloaded condition. The bolts are then inserted through the "channel iron" and bar to prevent bar rotation.

Following mounting and before Instron start up, some pre-test data is recorded (see appendix for test blank) information such as width, gage length, test temperature, etc.

Testing begins with the starting of the Instron and
the plotter at the same time. As the test progresses, ellipse dimensions, both major and minor axes, are measured and recorded on the plotter curve marking the location where the sampling was taken. Figure 3.9 show a typical MD oriented test with deformed ellipses. Width changes are also recorded in a similar manner to that of the eclipse data. Figure 3.10 shows typical test data. The onset of wave formation is noted by marking on the plotter curve as (Note: onset of wave formation was very loosely well. defined as that point in the test when out of plane deformation occured when all waves were no longer touching the grips. If the waves were still in contact with the bar then waves could be caused by a slack edge due to some minor misalignment.) Tests were taken either to failure or to two-thirds of the plotter's overall time range, then the Instron was stopped. If the sample had not failed, the plotter was allowed to run to its full time range, which recorded stress relaxation information on the sample.

Post-test data included final ellipse and minimum width dimensions in an attempt to measure elastic recovery after the sample is completely unloaded.

The following table shows number of tests run and their general location in the production line:

TABLE I

NUMBER OF TEST RUN

Material	Orientation		Number of	tests	
Polyester	[FP]	MD	4		
Polyester (300-D)	[FP]	MD	8		
Polyester (400-D)	[FP]	MD	2	2	
Polypropylene	[FP]	MD	22		
Polypropylene	[FP]	CD	22	22	
Polypropylene	[FP]	45	8	8	
Polypropylene	[TDO]	MD	30	30	
Polypropylene	[TDO]	CD	29		

FP : Final Product

TDO: Transverse Direction Orientation taken every ten feet down the length.



Figure 3.9 Polypropylene sample cut in the MD being tested at room temperature.



Figure 3.10 Typical time vs load plot for polypropylene in MD.

CHAPTER IV

EXPERIMENTAL RESULTS

Figures 4.1 - 4.3 show the MD specimen as it undergoes testing. Increasing length at a constant rate, the specimen deforms until failure. As it deforms, the test set up generates a load versus time plot, figure 4.4. The shape of the curve is typical for all of the tests that were run in the MD, regardless of the location of the sampling. The testing of the CD is shown in stages of deformation in figures 4.5 - 4.7. CD curves all have similar load/time curves as recorded in figure 4.8. Points were entered from the load/time curves into a computer program (see Appendix C) that converted the raw data into a more useful form. The beginning and ending points of the linear elastic regions were entered to establish Young's modulus for the test. Using the load and time points at which the test became plastic, the yield point was established. Points from the load/time plot were entered to determine the stress at specific strains. Neck dimension data was entered to yield stress information, as If the test specimen provided stress relaxation well. information, the appropriate load and time points were entered to yield relaxation time. After unloading or



Figure 4.1 Polypropylene sample with MD orientation awaiting loading.



Figure 4.2 Polypropylene sample with MD orientation under tension.



Figure 4.3 Polypropylene sample with MD orientation before failure.







Figure 4.5 Polypropylene sample with CD orientation before loading.



Figure 4.6 Polypropylene sample with CD orientation under tension.



Figure 4.7 Polypropylene sample with CD orientation before failure.



failure, final neck and ellipse dimensions were recorded. Using this data and the last neck or ellipse information, the program would yield elastic recovery. See Appendix C for the program's output from the above points.

Normalized moduli versus distance in feet from the center (figure 4.9) were plotted in bar chart form to show a profile of changes across the final product of polypropylene. The moduli were normalized with respect to thickness to aid comparison. Figure 4.10 is a plot of Emd/Ecd versus distance in feet from center. This plot of anistropy aids in the relationships between the MD and CD moduli. Plotting yield point, also normalized with respect to thickness, versus distance from the center in feet demostrates another anistropic aspect of the final product, figure 4.11. To understand how the final product came to be as it is, normalized moduli, MD and CD, were plotted down the length of the TDO. Emd/Ecd down the length of the TDO was also charted to help show how the biaxial process progressed (figures 4.12 and 4.13).

Dr. Good entered some of the information from one of the 45 degree tests into a finite element code and plotted stress distribution [27]. Figure 4.14 illustrates sigma x stresses in the 45 degree test while figure 4.15 shows the sigma y stresses. At the bottom grip, grid deformation was plotted in close up figure 4.16. An overall grid deformation was plotted in figure 4.17. All of the previous figures from the finite element model were



Figure 4.9 (E x t) vs distance from centerline of web. (Constant thickness of 0.79 mils)

NORMALIZED MODULUS







Distance from Center (ft)

Figure 4.11 Yield point vs distance from centerline of web. (Constant thickness of 0.79 mils)



Distance from Exit (ft)

Figure 4.12 Normalized modulus (E x t) vs position from exit of TDO (varying thickness).







Figure 4.14 Finite element model of 45 degree orientation of polypropylene featuring sigma x.



Figure 4.15 Finite element model of polypropylene oriented 45 degrees to MD. Featuring sigma y.



Figure 4.16 Grid deformation near grip for 45 degree orientation polypropylene sample modeled by finite element method.





generated in the elastic range.

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CHAPTER V

DISCUSSION

Static stress-strain tests on the final product of polypropylene were carried out on samples taken from different locations across the width of the web, starting from what was originally the center. From these tests, Young's moduli, normalized with respect to thickness, and plotted versus position from the center edge. In figure 4.9, the variations between normalized MD and CD moduli across the web, can be seen. The important point to note not only is that the MD and CD are not equal, but also that ratio of the modulus in the CD to that in the MD varies from 1.25 to 1.4. As can be seen in figure 4.10, the 1.25 ratio occurs near the center of the material while the 1.4 ratio of CD to MD appears at four feet from the center. The range of ratios may be caused by the distribution of stress as the polypropylene is oriented. Stress is applied in the desired direction of orientation in an attempt to align the molecular chains, but the stress distribution varies across the width of the web. This can be seen in the results of the finite element model, figure 4.14 [27]. If the stress varies across the web then the molecular alignment would also vary. Variations in molecular

alignment will result in variations in yield point and stiffness. Forty-five degree to MD tests corroborate the inequality of the MD and CD moduli, figure 5.1. Figure 5.1 shows a deviation of the major axis from the direction of applied tension. Because the stiffnesses in the MD and CD are unequal, tests run in the 45 degree orientation have principal stresses that are rotated away from the MD and CD directions of loading. This is indicated by the change in the ellipse's major and minor axis directions. This rotation of principal stress directions induce shear stresses in both the MD and CD directions that warp the printed grid and ellipse's major axis directions. This can be seen in figure 5.1 and the finite element model simulation, figure 4.15 [27]. The simulation uses the same defined conditions as the 45 degree test while still in the elastic region. Figure 4.15 [27] provides an indication of the local stresses involved. Figure 4.3, a test run in the MD orientation, shows the principal stresses remain in the same directions as MD and CD of loading according to the axes' directions. Equal stiffnesses in both MD and CD do not yield a deviation of principal stress directions like that in figure 5.1. The purpose of biaxial stretching includes achieving equal stiffness in MD and CD. Figure 4.10 illustrates the amount of anisotropy of stiffness versus distance from center. Notice the inclination of higher anisotropy as distance increases away from the center. Kimura and Shimizu [17] point out that within



Figure 5.1 Polypropylene sample with 45 degree orientation under tension

rectangular specimens "the maxima of stress and strain could be found at the center of the sample." This is due to boundary conditions and geometry. The boundary conditions are: negligible CD displacement at the grips and applied displacement in the MD. This and the results shown in figure 4.9 suggest the biaxial process works to a varying degree with respect to the width of the web in the tenter. The non-uniformity of the stress distribution is partially hampering the process.

Figures 4.14 - 4.17 [27] were plotted while still in the elastic region of deformation. As the web enters the plastic region, the maxima stress would approach the center. The chart in figure 4.11 shows the trend for yield point versus distance from the center edge is similar to the trend for stiffness. CD yield points are also 1.25 to 1.4 times higher than their MD counterparts. Because yield point and stiffness are similarly affected by the biaxial stretching, it is also affected by the uneven stress distribution instilled by the process in a likewise manner.

Tests were also run on samples taken from the treater down to almost half the length of the TDO. As across the final product, MD and CD normalized moduli varied. The CD moduli were almost always consistently higher than the MD. The TDO is theoretically where the CD stiffness is increased to match that of the MD in figure 4.12. It is possible that as the web was stretched and annealed in the MDO that the material's molecular alignment did not occur

to the degree desired. The MDO is supposed to achieve an MD preferred orientation, but the degree of orientation may not have been reached. It is possible the MDO stretching that occurs before the annealing takes place causes crazing that raises the internal energy of the material "due to new surface created inside the crazes" similar to results reported by Yefimov, Bulayev, Ozerin, Rebrov, Godovskii and Bakeyev [14]. This increase in internal energy could promote molecular chain randomness instead of alignment. As stated by Jenkins and Jenkins [10], there is an optimum temperature for achieve maximum preferred orientation. If that temperature is exceeded then the preferred orientation is decreased. Crazing could supply the additional energy needed to decrease the preferred orientation while in the Thicker samples, near the middle of the TDO, were MDO. visually observed to craze easily when loaded. The sample turned opaque in spots at a relative elongation of about 50 percent. As the test progressed, the opacity diminished.

Another possiblity that involves molecular alignment is a complex mixture of changing stress distribution, geometry and thermal energy. Because there was a relatively short time between the MDO operation and the start of the TDO stretching, little thermal energy has had time to be released. The geometry was changed between 70 and 40 feet and then held constant after 30 from the exit. If the molecular alignment was not given time to "set" after the stress from the TDO stretching was released, the web could have "snapped back" into its higher anisotropy configuration. Remember, the web is still at an elevated temperature and the TDO stress has been removed, but the inline tension that keeps the web on the rollers remains. This inline tension combined with stress relaxtion could cause the molecular chains to re-align in an unwanted manner. This might be prevented by allowing more "setting time" after the tenter.

Wave formation was also studied in both the polyester and the polypropylene. Wave formation is that point when the sample visibly displayed out of plane deformation. The onset of wave formation began within the elastic-plastic region described by Nilkanth [8] of the stress-strain curve usually before the yield point is reached. Typically, waves occurred at 80 percent of the yield point for the MD samples, but their formation in the CD direction behaved in a different manner. The onset of waves (CD direction tests) occurred at 125 percent of the yield values. Both MD and CD onsets were within the elastic-plastic region. This can be partially explained by looking at the elastic column buckling shown by Stevens [25] which is derived from Euler's formula for critical load. Only E and geometry play a part in that calculation for critical load. Therefore changes in E where the geometry is the same would proportionally affect the critical strain. The reverse is also true. Changing geometry, e.g. thickness, and keeping E constant would also change the point at which waves form.

Because CD is stiffer, it is presumed the critical strain would increase and therefore cause a higher strain for the onset of waves. However, the relationship between onset to wave formation and yield point is interesting. Euler's equation predicts "elastic" column buckling yet waves were observed to form in the elastic-plastic region. There seems to be some possible dependence on yield point. Euler's equation might be modified to include yield point for polymers to obtain a critical strain value for wave formation.

Mechanical tensile testing was chosen over optical and acoustical methods of determining E. They compare with the mechanical tensile testing in the following ways: Optical can measure displacements and Poisson's ratios but requires additional stress infomation to be useful (refer [15]); ultrasonics will return E values quickly from various points on the web, but those values are consistently higher than the mechanical E's observed. Mann, Baum, and Habeger [12] believe this is because ultrasonic testing gives much less time for viscoelastic relaxation than does mechanical tensile testing. Because there is not an established correlation between mechanically measured Young's moduli and acoustically measured values and the cost of the acoustical equipment is expensive, the mechanical tensile testing method was chosen.

Recommendations for future work include the following. It is necessary to find more accurate ways to

detect the onset of waves, such as through laser interferometry. More material based testing on paper and other polymers to derive a unitless number based on material properties and geometry for the onset of waves needs to be done. Very useful information can be collected by either scanning with a digitizing camera for use on a computer or taking numerous photographs with a scale in the picture. Thinning information will prove useful in wave formation considerations. Image analysis techniques, either with video frame grabbing or high resolution still photography could prove to be useful. Joint work between finite element modeling and material testing is required. Dymanmic testing that could be used on-line to measure characteristics of wave formation, such as the start of out of plane deformation, amplitude of deformation, stress state at formation, vibrational effects on viscoelasticity and their effect on permanent setting of waves. Temperature effects on wave formation and setting should also be studied. Direction and values of principle moduli need to be determined. Poisson's ratios and shear moduli are also needed.

CHAPTER VI

CONCLUS IONS

The biaxial orientation process used in production of this web works with a varying degree of success relative to the location of sampling. This can be seen in variations in yield point and stiffness throughout the web. These variations may be caused by unequal stress distribution that affects molecular alignment. It is also possible while the web was stretched and annealed in the MDO that alignment did not occur to the desired degree. Crazing may have played an important part here, raising the web's internal energy to a point where it adversely affected the molecular alignment. Along with this, insufficient "setting time" could have prevented the desired alignment. Further study of how the stress is distributed and how this distribution affects molecular alignment could prove invaluable in improving the properties of the web through production technique changes.

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APPENDIXES

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APPENDIX A

ANALYSIS OF UNCERTIANITY FOR NORMALIZED

YOUNG'S MODULUS

E*t=(F2-F1)*Lo/[(L2-L1)*W]

Where: Fl and F2 are the loads selected on the linear portion of the stress-strain curve; Ll and L2 are the lenghts of the elastically deformed specimen correponding with Fl and F2; Lo is the original length of the undeformed specimen; W is the width of the undeformed specimen.

Estimation of uncertainity intervals: $F=10 \pm 0.1$ lb $L=12 \pm 0.032$ in $W=6 \pm 0.032$ in

Relative Uncertainities:

 $\begin{array}{l} \text{Uf} = \pm (.1) / (10) = \pm 0.0100 \\ \text{U1} = \pm (0.032/12) = \pm 0.0027 \\ \text{Uw} = \pm (0.032/6) = \pm 0.0053 \\ \end{array} \\ \begin{array}{l} \text{U}_{\Delta} f = \pm \left[(0.0100)^2 + (0.0100)^2 \right]_{2}^{\frac{1}{2}} = \pm 0.0141 \\ \text{U}_{\Delta} 1 = \pm \left[(0.0027)^2 + (0.0027)^2 \right]_{2}^{\frac{1}{2}} = \pm 0.0038 \\ \end{array} \\ \begin{array}{l} \text{Ue} = \pm \left[(0.0141)^2 + (0.0038)^2 + (0.0053)^2 \right]_{2}^{\frac{1}{2}} = \pm 0.0155 \\ \text{or} \\ \text{Ue} = \pm 1.6 \text{ percent.} \end{array}$

APPENDIX B

ANALYSIS OF UNCERTAINITY FOR YIELD

POINT

Yp(pli)=F/W

Where: F is load at the yielding point and W is width of undeformed specimen.

from APPENDIX A:

 $Uf = \pm 0.0100$ $Uw = \pm 0.0027$

The relative uncertainity for Y is:

 $U_{y} = \pm |(Uf)^{2} + (Uw)^{2}|$ $U_{y} = \pm |(0.0100)^{2} + (0.0027)^{2}|^{\frac{1}{2}} = \pm 0.0100$

or Uy= ±1.00 percent.

APPENDIX C

COMPUTER PROGRAM FOR FINDING

ALL VALUES AND OUTPUT

t

```
t
1
                                  Mr. Thesis
1
Dim Neck$(50,3),E1c$(50,4),D$(100,25),E$(100,20,3),
    F$(100, 20, 4), Parml$(2), Parm2$(8)
Parml$(l)="Thickness (mm)"
Parml$(2)="Thickness (mils)"
Parm2$(1)=" Neck Reduction"
Parm2$(2)=" Ellipse A- circle A"
Parm2$(3)="
                  E (PSI)"
Parm2$(4)="Relaxation time (Sec)"
Parm2$(5)="Wave formation (PSI)"
Parm2$(6)="Wave formation (PLI)"
Parm2$(7)="Elastic Recovery"
Parm2$(8)=" Ell. Strains (True)"
5:
Cls
Print "Mr. Thesis"
Print "by Scott Robertson"
Print
Print "(D)ata entry"
Print "(C)alculation"
Print "(A)lter entry (correct)"
Print "(E)nd"
10:
A$=Upper$(Inkey$)
If A$="" Then
  Goto 10
Endif
A=Instr("DCAE",A$)
If A<>0 Then
  On A Gosub Zdata entry, Calc, Edit, Endit
Endif
Goto 5
End
1
```

```
Procedure Endit
  Cls
  Print "Are you sure you want to END (Y/N)?"
  15:
  A$=Upper$(Inkey$)
  If A$="" Then
    Goto 15
  Endif
  If A$="Y" Then
    End
  Endif
Return
Procedure Zdata entry
  If Exist("TEST.DAT") Then
    Open "A", #8, "TEST.DAT"
  Else
    Open "O", #8, "TEST.DAT"
  Endif
  20:
  Cls
  Print "(R)eturn to MAIN MENU"
  Print "(E)nter another test"
  25:
  A$=Upper$(Inkey$)
  If A$="" Then
    Goto 25
  Endif
  A=Instr("RE",A$)
  If A<>0 Then
    If A=1 Then
      Goto 30
    Endif
    If A=2 Then
      @Edata
      Goto 20
    Endif
  Endif
  Goto 25
  30:
  Close #8
Return
Procedure Edata
  40:
  @Input("Date",*Dated$)
  @Input("Test #",*Test$)
  @Input("Material note",*Material$)
  @Input("Thickness (mm)",*Thickness$)
  @Input("Specimen length (in.)",*Length$)
  @Input("Specimen width (in.)",*Width$)
  @Input("Material Orientation", *Material orientation$)
  @Input("Test temp. (F)",*Test temp$)
```

```
@Input("X scale (sec/cm)",*X scale$)
@Input("Y scale (v/cm)",*Y scale$)
@Input("E-Xl Coord. (in.)",*Exl$)
@Input("E-Yl_Coord. (in.)",*Eyl$)
@Input("E-X2 Coord. (in.)",*Ex2$)
@Input("E-Y2 Coord. (in.)",*Ey2$)
@Input("Wave formation x (in.)",*Wave x$)
@Input("Wave formation y (in.)",*Wave y$)
@Input("# of NECK data points",*Neck$)
W=Val(Neck$)
If W<>0 Then
  For X=1 To W
    @Input("Enter "+Str$(X)+" of "+Str$(W)+" NECK dim.
    (in.)",*B$)
    Neck(X,1) = B
     @Input("Enter "+Str$(X)+" of "+Str$(W)+" NECK X-
     coord. (in.)",*C$)
    Neck(X, 2) = C
    @Input("Enter "+Str$(X)+" of "+Str$(W)+" NECK Y-
    coord. (in.)",*D$)
    Neck(X,3)=D$
  Next X
Endif
@Input("# of ELC data points",*Elc$)
U=Val(Elc$)
If U<>0 Then
  For X=1 To U
    @Input("Enter "+Str$(X)+" of "+Str$(U)+" ELC Major
    dim. (in.)",*E$)
    Elc(X,1) = E
    @Input("Enter "+Str$(X)+" of "+Str$(U)+" ELC Minor
    dim. (in.)",*F$)
    Elc$(X,2) = F$
    @Input("Enter "+Str$(X)+" of "+Str$(U)+" ELC X-coord.
    (in.)",*G$)
    Elc$(X,3)=G$
    @Input("Enter "+Str$(X)+" of "+Str$(U)+" ELC Y-coord.
    (in.)",*H$)
    Elc$(X,4)=H$
  Next X
Endif
@Input("Start Stress Relax x-coord. (in.)",*Stressrx$)
@Input("Start Stress Relax y-coord. (in.)",*Stressry$)
@Input("2nd Stress Relax x-coord. (in.)",*Stressr2x$)
@Input("2nd Stress Relax y-coord. (in.)",*Stressr2y$)
@Input("Elastic Recovery Neck dim. (in.)",*Ernd$)
@Input("Elastic Recovery ELC Major dim. (in.)",
*Eremajord$)
@Input("Elastic Recovery ELC Minor dim. (in.)",
*Ereminord$)
Print
Print "Everything correct (Y/N)?"
50:
```

```
A$=Upper$(Inkey$)
  If A$="" Then
    Goto 50
  Endif
  If A$="N" Then
    Goto 40
  Endif
  If A$="Y" Then
    @Savedata
  Else
    Goto 50
  Endif
Return
Procedure Savedata
  @Output(Dated$)
  @Output(Test$)
  @Output(Material$)
  @Output(Thickness$)
  @Output(Length$)
  @Output(Width$)
  @Output(Material orientation$)
  @Output(Test temp$)
  @Output(X scale$)
  @Output(Y scale$)
  @Output(Ex1$)
  @Output(Eyl$)
  @Output(Ex2$)
  @Output(Ey2$)
  @Output(Wave x$)
  @Output(Wave y$)
  @Output(Neck\overline{\$})
  W=Val(Neck$)
  If W<>0 Then
    For X=1 To W
      @Output(Neck$(X,1))
      @Output(Neck$(X,2))
      @Output(Neck$(X,3))
    Next X
  Endif
  @Output(Elc$)
  U=Val(Elc$)
  If U<>0 Then
    For X=1 To U
       @Output(Elc$(X,1))
      @Output(Elc$(X,2))
      @Output(Elc$(X,3))
      @Output(Elc$(X,4))
    Next X
  Endif
  @Output(Stressrx$)
  @Output(Stressry$)
  @Output(Stressr2x$)
```

!18

!1 !2

13

14

15

16

17

18

19

110

111

112

113

!14

115

116

117

!19 !20 !21

```
@Output(Stressr2y$)
  @Output(Ernd$)
  @Output(Eremajord$)
  @Output(Ereminord$)
Return
.
Procedure Input(W$,X%)
  Print W$;
  Input Temp$
  If Temp$<>"" Then
    *X%=Temp$
  Endif
Return
1
Procedure Output(S$)
  If S$<>"" Then
    Print #8,S$
  Else
    Print #8, "None"
  Endif
Return
Procedure Calc
  Cls
  Print "Loading .... "
  Open "I", #8, "test.dat"
  Test=0
  Repeat
    Test=Test+1
    @Load(Test)
  Until Eof(#8)
  Close #8
  Print "Done."
  @Select
Return
Procedure Load(T)
  For Zz=1 To 17
    Input #8,D$(T,Zz)
  Next Zz
  If Val(D$(T,17)) <>0 Then
    For Zz=1 To Val(D$(T,17))
      For W=1 To 3
        Input #8, E$(T, Zz, W)
      Next W
    Next Zz
  Endif
  Input #8,D$(T,18)
  If Val(D$(T,18)) <>0 Then
    For Zz=1 To Val(D$(T,18))
      For W=1 To 4
         Input #8,F$(T,Zz,W)
      Next W
```

1	22	
1	23	
!	24	

125

```
Next Zz
  Endif
  For Zz=19 To 25
    Input #8,D (T,Zz)
  Next Zz
Return
Procedure Decimal(A$,X%)
  Let Decimal=0
  If Instr(A$,".")=0 And Instr(A$,"/")<>0 Then
    A=Instr(A$," ")
    If A<>0 Then
      Let Decimal=Val(Left$(A$,A))
      A$=Mid$(A$,A+1)
    Endif
    C=Instr(A$,"/")
    If C<>0 Then
      Let Decimal=Decimal+Val(Left$(A$,C))/Val(Mid$(A$,C+1)
      )
    Endif
  Else
    Let Decimal=Val(A$)
  Endif
  *X%=Decimal
Return
Procedure E
  Cdo(D(T,4)+D(T,5)+D(T,6)+D(T,11)+D(T,12)+D(T,13))
  +D$(T,14))
  If Cdo! Then
    @Decimal(D$(T,5),*Lo)
    (Decimal(D$(T,6),*W))
    Ll=QXcon(Val(D$(T,11)),T)
    L2 = QX con(Val(D$(T, 13)), T)
    @Force(D$(T,12),*Forcel)
    @Force(D$(T,14),*Force2)
    Thickness=Val(D$(T,4))/25.4
    Deltal=L2-L1
    E=(Force2-Forcel)*Lo/(Deltal*W*Thickness)
    Round=Int(Log10(E+1))+1
    Mid$(Work$,67,Round)=Str$(E)
  Else
    Mid$ (Work$, 67, 4) = "NCDO"
  Endif
Return
Deffn Tcon(T)=T/25.4
Deffn Xcon(X,T) = Val(D$(T,9))*0.0212*X
Procedure E area
  On Which Gosub Screen, Prnter
  Work$=K$
  Mid$(Work$,40,19)="Strain Stress(psi)"
```

```
On Which Gosub Screen, Prnter
  Kr = Val(D\$(T, 18))
  If Kr<>0 Then
    For D=1 To Kr
      Work$=K$
      @Cdo(D$(T,11)+F$(T,D,1)+F$(T,D,2)+F$(T,D,3)+F$(T,D,4
      ))
  If Cdo! Then
        (Decimal(D$(T,5),*Lo))
        @Decimal(F$(T,D,1),*Major)
        @Decimal(F$(T,D,2),*Minor)
        (Decimal(D$(T,6),*W))
        E area=Pi*Major*Minor/4-Pi/4
        X_j=Q_{con}(V_a1(F_{(T,D,3)}),T)-Q_{con}(V_a1(D_{(T,11)}),T)
        Xj=Xj/Lo
        QStress(W,FS(T,D,4),*Yj)
        Mid\$(Work\$, 40, 5) = Left\$(Str\$(Xj), 5)
        Mid$(Work$,50,Int(Log10(Yj+1)+1))=Str$(Yj)
        Mid$(Work$,67,5)=Left$(Str$(E area),5)
      Else
        Mid$ (Work$, 40, 4) = "NCDO"
      Endif
      On Which Gosub Screen, Prnter
    Next D
  Endif
  Work$=K$
Return
Procedure Cdo(Check$)
  Cdo!=True
  If Instr(Upper$(Check$),"NONE")<>0 Then
    Cdo!=False
  Endif
Return
1
Procedure Select
  Select2:
  Cls
  Print "(1): Select print parms"
  Print "(2): Test order"
  Print "(3): Print"
  Print "(4): Quit"
  Print
  @One key(4)
  If Key=4 Then
    Goto Selectl
  Endif
  On Key Gosub Parms, Order, Prnt
  Goto Select2
  Selectl:
Return
```

```
Procedure One key(O)
  Key:
  A$=""
  A$=Inkey$
  If A$="" Then
    Goto Key
  Endif
  If Val(A$)>=1 And Val(A$)<=0 Then
    Key=Val(A$)
  Else
    Goto Key
  Endif
Return
1
Procedure Order
  Order$=""
  Order:
  Cls
  Print "Current Order: ";Order$
  Input "Enter test# (Return/returns, All/All)";J$
  If J$="" Then
    Goto Order2
  Endif
  If Upper$(J$)="ALL" Then'
    For Dd=1 To Test
       @Order sum(D$(Dd,2))
    Next Dd
    Goto Order2
  Endif
  @Order sum(J$)
  Goto Order
  Order2:
Return
Procedure Order sum(H$)
  If Order$="" Then
    Order$=H$
  Else
    Order$=Order$+","+H$
  Endif
Return
1
Procedure Parms
  Cls
  Print "Select Parm 1"
  Print "(1): Thickness (mm)"
  Print "(2): Thickness (mils)"
  @One key(2)
  Pl=Key
  Print
  Print "Select Parm 2"
  Print "(1): Neck Reduction"
  Print "(2): Ellipse area- 1 inch. circle area"
```

```
Print "(3): E (PSI)"
  Print "(4): Relaxation time (Sec)"
  Print "(5): Wave formation (PSI)"
  Print "(6): Wave formation (PLI)"
  Print "(7): Elastic Recovery"
  Print "(8): Ell. Strains (Engr)"
  @One key(8)
  P2=Key
  Print
  Print "Select Parm 3"
  Print "(1): Pre 4/14/88 correction factor"
  Print "(2): Exit"
  One key(2)
  P3 = -(Key = 1) * 0.35 - (Key = 2)
Return
Procedure Prnt
  Cls
  K$=String$(79," ")
  Print "(1): Screen"
  Print "(2): Printer"
  @One key(2)
  Whic\overline{h}=Key
  Print "Want a header? (Y/N)"
  Prnt:
  A$=""
   A$=Inkey$
  If A$="" Then
    Goto Prnt
  Endif
  A$=Upper$(A$)
  If A$<>"Y" And A$<>"N" Then
    Goto Prnt
  Endif
  If Which=2 Then
    Open "", #7, "PRN:"
  Endif
  If A$="Y" Then
    Header$=K$
    First=Len(Parml$(Pl))
    Last=Len(Parm2$(P2))
    Mid$(Header$,42,First)=Parml$(Pl)
    Mid$(Header$,59,Last)=Parm2$(P2)
    Mid$(Header$,1,6)="Test #"
    Mid$(Header$,6,14)="Mat. Orient."
    Mid$(Header$,22,9)="Mat. Note"
    Work$=Header$
    On Which Gosub Screen, Prnter
    Work$=String$(79,"-")
    On Which Gosub Screen, Prnter
  Endif
  Repeat
    Work$=K$
```

```
Xcv=Val(Order$)
    T=0
    Repeat
      T=T+1
    Until (Xcv=Val(D$(T,2)) Or T=>Test)
    Ww=Instr(Order$,",")
    Mid$ (Work$, 1, Len (D$ (T, 2)) = D$ (T, 2)
    Mid(Work, 6, Len (D(T, 7)))=D(T, 7)
    Mid$(Work$,11,Len(D$(T,3)))=D$(T,3)
    On Pl Gosub Thickmm, Thickmils
    On P2 Gosub Necking, E area, E, Relax, Wavepsi, Wavepli,
    Elastic,Ed
    On Key Gosub Screen, Prnter
    If Ww<>0 Then
      Order$=Mid$(Order$,Ww+1)
    Endif
  Until Ww=0
  If Which=2 Then
    Close #7
  Endif
  Input "Hit return";Ab$
Return
Procedure Screen
  Print Work$
Return
Procedure Prnter
  Print #7,Work$
Return
Procedure Thickmm
  Mid$ (Work$, 48, Len(D$(T, 4)))=D$(T, 4)
Return
Procedure Thickmils
  Mils=1000*Val(D$(T,4))/25.4
  Mid$(Work$,48,5)=Left$(Str$(Mils),5)
Return
Procedure Relax
  QCdo(D$(T,19)+D$(T,20)+D$(T,21)+D$(T,22))
  If Cdo! Then
    Zzz=2.54*Val(D$(T,9))
    Timl=Zzz*Val(D$(T,19))
    Tim2=Zzz*Val(D$(T,21))
    (Decimal(D$(T,6),*W))
    Relax = -(Tim2-Tim1) / (Log(Val(D$(T,22))/Val(D$(T,20))))
    Mid (Work $, 67, 5) = Left $ (Str $ (Relax), 5)
  Else
    Mid$ (Work$,67,4) = "NCDO"
  Endif
Return
```

```
Procedure Wavepsi
  QCdo(D$(T,11)+D$(T,15)+D$(T,16))
  If Cdo! Then
    (Decimal(D$(T,6),*W))
    @Stress(W,D$(T,16),*Stress)
    Mid$(Work$,67,Int(Log10(Stress+1)+1))=Str$(Stress)
  Else
    Mid$ (Work$,67,4) = "NCDO"
  Endif
Return
Procedure Wavepli
  (Cdo(D(T,11)+D(T,15)+D(T,16)))
  If Cdo! Then
    (Decimal(D$(T,6),*W))
    @Force(D$(T,16),*Pli)
    Pli=Pli/W
    Mid$(Work$,67,5)=Left$(Str$(Pli),5)
  Else
    Mid$ (Work$,67,4) = "NCDO"
  Endif
Return
Procedure Necking
  On Which Gosub Screen, Prnter
  Work$=K$
  Mid$(Work$,10,21)="TStrain TStress(psi)"
  Mid$(Work$,40,21)="EStrain EStress(psi)"
  On Which Gosub Screen, Prnter
  Kr = Val(D\$(T, 17))
  If Kr<>0 Then
    For D=1 To Kr
      Work$=K$
      (Cdo(D$(T,11)+E$(T,D,1)+E$(T,D,2)+E$(T,D,3))
      If Cdo! Then
        (Decimal(D$(T,5),*Lo))
        (Decimal(D$(T,6),*W))
        @Decimal(E$(T,D,1),*Necko)
        Neck = (W-Necko) / W
        X_{j}=Q_{con}(V_{al}(E_{(T,D,2)},T)-Q_{con}(V_{al}(D_{(T,11)}),T)
        Xtrue=Log((Xj+Lo)/Lo)
        Xj=Xj/Lo
        @Force(E$(T,D,3),*Ytrue)
        Ytrue=25.4*Ytrue/(Necko*Val(D$(T,4)))
        @Stress(W,E$(T,D,3),*Yj)
        Mid$(Work$,10,5)=Left$(Str$(Xtrue),5)
        Mid$(Work$,20,Int(Log10(Ytrue+1)+1))=Str$(Ytrue)
        Mid$(Work$,40,5)=Left$(Str$(Xj),5)
        Mid$(Work$,50,Int(Log10(Yj+1)+1))=Str$(Yj)
        Mid (Work $, 67, 5) = Left $ (Str $ (Neck), 5)
      Else
        Mid$ (Work$, 40, 4) = "NCDO"
```

Endif On Which Gosub Screen, Prnter Next D Endif Work\$=K\$ Return Procedure Stress(W,H\$,X%) @Force(H\$,*F) *X%=25.4*F/(W*Val(D\$(T,4))) Return . Procedure Force (G\$,F%) Volts=Val(G\$) *Val(D\$(T,10)) If Val(D\$(T,2))>44 Then *F%=P3*Volts*356 Else *F%=P3*Volts*45.9 Endif Return Procedure Edit Cls Print "Loading...." Open "I",#8,"test.dat" Test=0 Repeat Test=Test+1 @Load(Test) Until Eof(#8) Close #8 Print "Done." Ditl: Input "Test# to edit (return to quit)";T\$ If T\$<>"" Then Gt=0Repeat Gt=Gt+1 Until (Val(T\$)=Val(D\$(Gt,2)) Or Gt>=Test) @Display(Gt) Goto Ditl Endif Open "O", #8, "Edited.dat" For T=1 To Test For Zz=1 To 17 Print #8,D\$(T,Zz) Next Zz If Val(D\$(T,17)) <>0 Then For Zz=1 To Val(D\$(T,17))For W=1 To 3 Print #8,E\$(T,Zz,W) Next W Next Zz

```
Endif
    Print #8,D$(T,18)
    If Val(D$(T,18)) <>0 Then
      For Zz=1 To Val(D$(T,18))
        For W=1 To 4
          Print #8,F$(T,Zz,W)
        Next W
      Next Zz
    Endif
    For Zz=19 To 25
      Print #8,D$(T,Zz)
    Next Zz
  Next T
  Close #8
  Kill "test.dat"
  Name "edited.dat" As "test.dat"
Return
Procedure Display(T)
  Dis:
  Cls
  For Zz=1 To 17
    Print Zz;":";D$(T,Zz),
  Next Zz
  If Val(D$(T,17)) <>0 Then
    For Zz=1 To Val(D$(T,17))
      For W=1 To 3
        Print "N ";Zz;",";W;":";E$(T,Zz,W),
      Next W
    Next Zz
  Endif
  Print 18;":";D$(T,18),
  If Val(D$(T,18)) <>0 Then
    For Zz=1 To Val(D$(T,18))
      For W=1 To 4
        Print "E ";Zz;",";W;":";F$(T,Zz,W),
      Next W
    Next Zz
  Endif
  For Zz=19 To 25
    Print Zz;":";D$(T,Zz),
  Next Zz
  Print
  Input "Enter character";A$
  A$=Upper$(A$)
  Input "# 1";Dx$
  Dx=Val(Dx$)
  Input "# 2 ";Dy$
  Dy=Val(Dy$)
  If Val(A$) >= 1 And Val(A$) <= 25 Then
    Input "Enter new value";D$(T,Val(A$))
    Goto Dis
  Endif
```

```
If A$="N" Then
    Input "Enter new neck";E$(T,Dx,Dy)
    Goto Dis
  Endif
  If A$="" Then
    Goto Dis3
  Endif
  If A$="E" Then
    Input "Enter new neck";F$(T,Dx,Dy)
    Goto Dis
  Endif
  Dis3:
Return
Procedure Elastic
  On Which Gosub Screen, Prnter
  Work $=K$
  Mid$(Work$,10,33)="Elastic Neck Recover% (Nu-N1/N1)="
  (O(D)(T, 23))
  If Cdo! Then
    Rt=Val(D$(T,17))
    If Rt<>0 Then
      @Decimal(E$(T,Rt,l),*Nl)
      @Decimal(D$(T,23),*Nu)
      Hk = (Nu - Nl) / Nl
      Mid\$(Work\$, 51, 5) = Left\$(Str\$(Hk), 5)
    Else
      Mid(Work, 51, 4) = "NCDO"
    Endif
  Else
    Mid\$(Work\$, 51, 4) = "NCDO"
  Endif
  On Which Gosub Screen, Prnter
  Work$=K$
  Mid$(Work$,10,39)="Elastic Major Axis Recover% (Mu-
  M1/M1) = "
  (D$(T, 24))
  If Cdo! Then
    Rt=Val(D$(T,18))
    If Rt<>0 Then
      @Decimal(F$(T,Rt,l),*Nl)
      (Decimal(D$(T,24),*Nu))
      Hk = (Nu - Nl) / Nl
      Mid$ (Work$,51,5)=Left$ (Str$(Hk),5)
    Else
      Mid\$(Work\$, 51, 4) = "NCDO"
    Endif
  Else
    Mid$ (Work$, 51, 4) = "NCDO"
  Endif
      On Which Gosub Screen, Prnter
  Work$=K$
  Mid$(Work$,10,39)="Elastic Minor Axis Recover% (Mu-
```

```
Ml/Ml) = "
      @Cdo(D$(T,25))
      If Cdo! Then
        Rt = Val(D\$(T, 18))
    If Rt<>0 Then
      (Ps(T,Rt,2),*N1)
      @Decimal(D$(T,25),*Nu)
      Hk = (Nu - Nl) / Nl
      Mid$ (Work$,51,5)=Left$ (Str$(Hk),5)
    Else
      Mid(Work, 51, 4) = "NCDO"
    Endif
  Else
    Mid$ (Work$, 51, 4) = "NCDO"
  Endif
  On Which Gosub Screen, Prnter
  Work$=K$
Return
Procedure Ed
  On Which Gosub Screen, Prnter
  Work$=K$
  Mid$(Work$,40,37)="Strain Stress(psi)
                                               Major
                                                         Minor"
  On Which Gosub Screen, Prnter
  Kr = Val(D\$(T, 18))
  If Kr<>0 Then
    For D=1 To Kr
      Work$=K$
      (0Cdo (D(T,11)+F(T,D,1)+F(T,D,2)+F(T,D,3)+F(T,D,4))
      ))
      If Cdo! Then
        @Decimal(F$(T,D,1),*Major)
        @Decimal(F$(T,D,2),*Minor)
        (Decimal(D$(T,5),*Lo))
        (Decimal(D$(T,6),*W))
        X_j=Q_xcon(Val(F_s(T,D,3)),T)-Q_xcon(Val(D_s(T,11)),T)
        Xj=Xj/Lo
        QStress(W,F$(T,D,4),*Yj)
        Mid$ (Work$, 40, 5) = Left$ (Str(X_j), 5)
        Mid$ (Work$,50, Int(Log10(Yj+1)+1))=Str$(Yj)
        Ma=Int(Log(Major)*10000)/10000
        Mi=Int(Log(Minor)*10000)/10000
        Mid$ (Work$,63,5)=Left$ (Str$ (Ma),5)
        Mid$ (Work$,72,5)=Left$ (Str$ (Mi),5)
        If Abs(Abs(Ma)-Abs(Mi))>0.06 Then
           Mid$(Work$,78,1)="*"
        Endif
      Else
        Mid$ (Work$, 40, 4) = "NCDO"
      Endif
      On Which Gosub Screen, Prnter
    Next D
  Endif
  Work$=K$
Return
```

Tes	st Mat.Or	ient. Mat	OUTPUT	Thick.	(mils)	Neck Reduc.
 31 7	MD PP S Strain TSt	outh Piece ress(psi)	EStra	0.787 in ESt	ress(psi)
35	MD PP S TStrain TS	outh Piece tress(psi)	EStra	0.787 in ESt	ress(psi)
40	CD PP S TStrain TS 0.087 0.135	outh Piece tress(psi) 8617 13248	EStr 0.09 0.14	0.787 ain ES 1 8 4 1	578 tress(ps 433 2402	0.125 0.021 0.063
42	0.200 CD PP S	19431 South Piece	0.22	1 1 0.787	7363	0.106
30	CD PP N	11906 Inrth Piece	EStr 0.1	36	tress(ps 11420	0.040
50	TStrain TS 0.091	stress(psi) 3442	EStr 0.0	ain ES 95	tress(ps 3285	i) 0.045
57	MD PP T TStrain TS 0.107 0.205	DO 0' from tress(psi) 3752 5161	Exit EStr 0.1 0.2	0.866 ain ES 13 27	tress(ps 3596 4623	i) 0.041 0.104
62	MD PP T TStrain TS 0.100 0.194 0.280	DO 0' from tress(psi) 4288 5978 7618	Exit EStr 0.1 0.2 0.3	0.866 ain ES 06 14 23	stress(ps 4110 5480 6507	0.041 0.083 0.145
61	MD PP T TStrain TS 0.091	DO 0' from tress(psi) 4697	Exit EStr 0.0	0.866 ain ES 95	tress(ps 4513	i) 0.039
53	CD PP T TStrain TS 0.071	DO 0' from tress(psi) 9932	Exit EStr 0.0	0.866 ain ES 74	tress(ps 9535	i) 0.04
60	CD PP T TStrain TS 0.105	DO 0' from Stress(psi) 11437	Exit EStr 0.1	0.866 ain ES 11	Stress(ps 10960	si) 0.041
55	MD PP T TStrain TS 0.093 0.190	DO 20' from Stress(psi) 3366 4623	exit EStr 0.0 0.2	1.259 ain ES 198 209	Stress(ps 3229 4152	si) 0.040 0.102

59 MD PP TDO 20' from exit 1.259

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	TStrain 0.112 0.176	TStress(psi) 3516 4205	EStrain 0.119 0.193	EStress(psi) 3296 3767	0.062 0.104
91	CD P TStrain 0.077 0.118	P TDO 20' from TStress(psi) 4997 7125	exit 1.25 EStrain 0.080 0.125	59 EStress(psi) 4945 6828	0.010 0.041
90	MD P TStrain 0.085 0.128 0.187 0.241 0.301 0.343 0.410	P TDO 30' from TStress(psi) 2319 2583 3032 3523 3936 4410 5861	exit 1.3 EStrain 0.089 0.136 0.206 0.273 0.351 0.410 0.507	77 EStress(psi) 2251 2455 2762 3069 3274 3581 3888	0.029 0.049 0.089 0.128 0.168 0.188 0.336
72	MD P TStrain 0.105 0.170 0.235 0.336	P TDO 30' from TStress(psi) 2741 3298 3789 4551	exit 1.3 EStrain 0.110 0.185 0.265 0.399	77 EStress(psi) 2583 2981 3279 3676	0.057 0.096 0.134 0.192
85	MD P TStrain 0.100	P TDO 40' from TStress(psi) 2172	exit 1.4 EStrain 0.106	56 EStress(psi) 2125	0.021
87	MD F TStrain 0.100 0.152 0.192 0.213 0.259 0.292 0.345	PP TDO 40' from TStress(psi) 2160 2606 2888 3069 3258 3620 4010	exit 1.4 EStrain 0.106 0.164 0.212 0.238 0.296 0.339 0.413	56 EStress(psi) 2094 2394 2593 2693 2793 2992 3192	0.030 0.081 0.102 0.122 0.142 0.173 0.204
83	CD F TStrain 0.113	PP TDO 40' from TStress(psi) 5430	exit 1.4 EStrain 0.119	56 EStress(psi) 5199	0.042
77	CD F TStrain 0.085 0.115 0.159	P TDO 40' from TStress(psi) 5332 7047 8970	exit 1.4 EStrain 0.089 0.122 0.172	56 EStress(psi) 5100 6587 8287	0.043 0.065 0.076
80	MD F TStrain 0.115	PP TDO 50' from TStress(psi) 2511	exit 1.5 EStrain 0.122	574 EStress(psi) 2354	0.062

	0.168 0.246 0.301 0.327	2877 3639 4173 4462	0.184 0.278 0.351 0.387	2637 3108 3390 3579	0.083 0.145 0.187 0.197
79	MD TStrain 0.115	PP TDO 50' from TStress(psi) 2441	exit l. EStrain 0.122	574 EStress(psi) 2347	0.038
33	MD TStrain 0.157 0.231 0.338 0.306	PP TDO 60' from TStress(psi) 1118 1369 1833 2214	exit 1. EStrain 0.171 0.260 0.402 0.358	771 EStress(psi) 1025 1198 1489 1684	0.083 0.125 0.187 0.239
43	MD TStrain 0.141 0.244 0.331 0.377 0.435	PP TDO 60' from TStress(psi) 3051 4194 5314 6021 6764	exit 1. EStrain 0.151 0.277 0.393 0.459 0.545	771 EStress(psi) 2860 3670 4318 4641 5073	0.062 0.125 0.187 0.229 0.25
50	MD TStrain 0.071 0.187	PP TDO 60' from TStress(psi) 2992 5297	exit 1. EStrain 0.074 0.206	771 EStress(psi) 2930 4856	0.020
65	MD TStrain 0.211 0.246 0.324 0.373	PP TDO 60' from TStress(psi) 54477 54207 65673 74194	exit l. EStrain 0.235 0.279 0.382 0.453	771 EStress(psi) 50030 47569 54951 59052	0.081 0.122 0.163 0.204
52	CD TStrain 0.059	PP TDO 60' from TStress(psi) 3444	exit l. ⁷ EStrain 0.060	771 EStress(psi) 3364	0.023
78	CD TStrain 0.077	PP TDO 60' from TStress(psi) 3253	exit l. EStrain 0.080	771 EStress(psi) 3105	0.045
69	CD TStrain 0.113 0.192 0.260 0.300 0.351 0.393	PP TDO 70' from TStress(psi) 4438 6200 7595 8346 9580 10166	exit 1.9 EStrain 0.120 0.212 0.297 0.350 0.421 0.482	968 EStress(psi) 4245 5661 6604 7076 7705 8177	0.043 0.086 0.130 0.152 0.195 0.195

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105 MD #1 (6" from center TStrain TStress(psi) 0.166 3741	edge) 0.787 EStrain EStress(psi) 0.181 3426	0.084
103 MD #2 (12" from center TStrain TStress(psi) 0.070 3078 0.159 4420 0.219 5046	edge) 0.787 EStrain EStress(psi) 0.072 3014 0.172 4144 0.245 4521	0.020 0.062 0.104
<pre>109 MD #5 (42" from center TStrain TStress(psi) 0.077 3078 0.130 3974 0.178 4521 0.219 5046 0.276 5734 0.317 6409 0.359 7138 0.410 7927</pre>	edge) 0.787 EStrain EStress(psi) 0.080 3014 0.139 3767 0.195 4144 0.245 4521 0.318 4897 0.373 5274 0.432 5651 0.507 6028	0.020 0.052 0.083 0.104 0.145 0.177 0.208 0.239
<pre>108 MD #6 (48" from center TStrain TStress(psi) 0.115 3889</pre>	edge) 0.787 EStrain EStress(psi) 0.122 3728	0.041
115 MD #10 (48" from center TStrain TStress(psi) 0.035 2260	edge) 0.787 EStrain EStress(psi) 0.036 2214	0.020
<pre>114 MD #11 (96" from center TStrain TStress(psi) 0.038 2474 0.103 3538 0.168 4521 0.201 4893 0.250 5513</pre>	edge) 0.787 EStrain EStress(psi) 0.039 2448 0.108 3390 0.184 4144 0.223 4332 0.284 4709	0.010 0.041 0.083 0.114 0.145
97 MD #12(102" fromm cente TStrain TStress(psi) 0.056 2637 0.123 3734	r edge) 0.787 EStrain EStress(psi) 0.058 2583 0.131 3506	0.020 0.061
100 CD #13 (6" from cente TStrain TStress(psi) 0.054 5194 0.100 9042	r edge) 0.787 EStrain EStress(psi) 0.055 5086 0.106 8665	0.020 0.041
104 CD #14 (6" from cente TStrain TStress(psi) 0.088 8850	r edge) 0.787 EStrain EStress(psi) 0.092 8665	0.020
93 CD #16 (42" from cente TStrain TStress(psi)	r edge) 0.787 EStrain EStress(psi)	

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	0.100 0.164	9435 13563		0.106 0.178	9042 12433	0.041 0.083
113	CD # TStrain 0.027 0.090 0.120	17 (78" fr TStress(3078 7551 10363	om cer psi)	nter edge) 0. EStrain 0.027 0.094 0.128	787 EStress(psi 3045 7233 9708) 0.010 0.042 0.063
101	CD # TStrain 0.075 0.115 0.149	18 (78" fr TStress(6059 8465 11008	om cer psi)	nter edge) 0. EStrain 0.078 0.122 0.161	787 EStress(psi 5845 8037 10229) 0.035 0.050 0.070
96	CD # TStrain 0.056 0.100	19 (84" fr TStress(5274 8272	om cer psi)	nter edge) 0. EStrain 0.058 0.106	787 EStress(psi 5167 7935) 0.020 0.040
66	CD # TStrain 0.149 0.192	20 (84" fr TStress(13664 17664	om cer psi)	nter edge) 0. EStrain 0.161 0.212	787 EStress(psi 12810 15824) 0.062 0.104
116	MD TStrain 0.125 0.159 0.232 0.288 0.301 0.301	PP TDO 10 TStress(1048 1205 1435 1617 1786 1786	from psi)	exit 1. EStrain 0.133 0.172 0.262 0.334 0.351 0.351	181 EStress(psi 994 1118 1242 1367 1491 1491) 0.051 0.072 0.134 0.154 0.164 0.164
117	45 TStrain 0.082 0.141 0.262	PP North F TStress(2358 3269 4973	iece psi)	0 [.] EStrain 0.085 0.151 0.300	787 EStress(psi 2308 3078 4232) 0.021 0.058 0.148
118	CD TStrain 0.106 0.162	PP TDO 10' TStress(2157 2883	from psi)	exit 1. EStrain 0.111 0.176	181 EStress(psi 2049 2652) 0.05 0.08
119	45 TStrain 0.107 0.171 0.233	PP North F TStress(1808 2944 4410	Piece (psi)	0. EStrain 0.113 0.186 0.262	787 EStress(psi 1695 2637 3767) 0.062 0.104 0.145
121	MD TStrain	PP TDO 10 TStress(from (psi)	exit l. EStrain	.181 EStress(psi)

(0.002		0 0			0.002	2 (5 (0 0	0.02
Mat. Reduc TStra EStra EStra Neck	: Mat c.: R ain: ess: ain: ess: Redu	eria educ True True Engi Engi c.:	l, Or tion, Stra Stre neeri neeri (Orig width	ient PP: in w ss w ng S ng S inal	Polyp vith re vith re train tress width	entatic ropyler spect t spect t with re with re of web	on, Th ne. to neck to neck espect espect o-New	ick.: Th k. to neck to neck width)/O	ickness, riginal
Test	Mat.	Ori	ent.	Mat.	Note	Thick	. (mil:	s) Ell.	A-circle
31	MD	PP	South	Pie S C C	ece Strain 0.189 0.277 0.487	0.7 Stress 1530 1748 213	787 s(psi)) 3 7		-0.01 -0.08 -0.04
35	MD	PP	South	Pi€ 2 0 0	ece Strain 0.130 0.226	0.3 Stress 1263 1509	787 s(psi) 3 5		-0.05 -0.02
40	CD	PP	South	Pie S	ece Strain).110	0. Stress 9922	787 s(psi) 2		-0.05
42	CD	PP	South	Pie S	ece Strain 0.103	0.3 Stress 8569	787 s(psi) 5		-0.05
26	MD	PP	North	Pie S (ece Strain 0.111 0.209	0. Stres: 558 704	787 s(psi) 6 3		-0.00
57	MD	PP	TDO 0	' fr S	om Exi Strain).206	t 0.8 Stress 4453	366 s(psi) 2		-0.01
62	MD	PP	TDO 0	' fr S (com Exi Strain).143).275	t 0.8 Stress 479 616	366 s(psi) 5 5		-0.05 0.030
60	CD	PP	TDO 0	' fr 9 (com Exi Strain).143	t 0.8 Stres: 1370	866 s(psi) 00		-0.05
55	MD	PP	TDO 2	0' f	from ex	it 1.3	259		

			Strain Stress(psi) 0.151 3690	-0.05
59	MD	PP TDO 20'	from exit 1.259 Strain Stress(psi) 0.151 3532	-0.01
91	CD	PP TDO 20'	from exit 1.259 Strain Stress(psi) 0.064 4003 0.111 6357	-0.02 -0.05
94	ĊD	PP TDO 20'	from exit 1.259 Strain Stress(psi) 0.084 4914	-0.05
90	MD	PP TDO 30'	from exit 1.377 Strain Stress(psi) 0.117 2353 0.184 2660 0.239 2967 0.306 3171 0.390 3478 0.471 3785	-0.05 -0.03 0.052 0.058 0.024 -0.00
72	MD	PP TDO 30'	from exit 1.377 Strain Stress(psi) 0.096 2484 0.202 2981 0.334 3577	-0.09 -0.06 -0.04
85	MD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.089 2018 0.167 2337 0.217 2762	-0.05 -0.03 -0.06
87	MD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.090 1995 0.196 2493 0.304 2793 0.402 3192	-0.07 -0.08 -0.08 -0.07
83	CD	PP TDO 40'	from exit l.456 Strain Stress(psi) 0.103 4679	-0.02
77	CD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.106 5950 0.156 7969	-0.07 -0.05

80	MD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.100 2166 0.248 3014 0.384 3579	-0.07 -0.04 -0.04
79	MD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.114 2260	-0.07
25	CD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.193 10201 0.251 11901	-0.01 -0.02
33	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.154 982 0.330 1360 0.450 1586	-0.01 -0.11 0.024
43	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.111 2536 0.217 3346 0.353 4102 0.456 4318 0.512 4911 0.585 5289	-0.15 -0.06 -0.04 -0.01 -0.01 -0.00
50	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.132 3935 0.307 5525	-0.05 0.012
65	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.135 34447 0.235 44289 0.412 57412	-0.05 -0.06 -0.04
52	CD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.166 7009	-0.05
69	CD	PP TDO 70'	from exit 1.968 Strain Stress(psi) 0.103 3774 0.191 5346 0.279 6447 0.359 7233 0.432 7862	-0.09 -0.01 -0.02 0.024 0.171

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105 MD #1(6" from	center edge) 0.787 Strain Stress(psi) 0.111 3045	-0.02
103 MD #2(12" from	center edge) 0.787 Strain Stress(psi) 0.055 3014 0.114 3579 0.206 4144	-0.02 -0.05 -0.06
106 MD #4(30" from	center edge) 0.787 Strain Stress(psi) 0.055 3045 0.122 3807	-0.02 -0.05
109 MD #5(42" from	center edge) 0.787 Strain Stress(psi) 0.066 3014 0.122 3579 0.178 4144 0.228 4521 0.301 4897 0.357 5274 0.418 5651 0.490 6028	$ \begin{array}{c} -0.05 \\ -0.05 \\ -0.06 \\ -0.05 \\ -0.03 \\ -0.14 \\ -0.04 \\ \end{array} $
108 MD #6(48" from	center edge) 0.787 Strain Stress(psi) 0.066 2983 0.106 3355 0.172 4101	-0.02 -0.02 -0.03
115 MD #10(48" from	n center edge) 0.787 Strain Stress(psi) 0.061 2583	-0.00
114 MD #11(96" from	n center edge) 0.787 Strain Stress(psi) 0.061 2825 0.150 3767 0.251 4521 0.256 4521	-0.02 -0.00 0.009 0.001
97 MD #12(102" form	n center edge) 0.787 Strain Stress(psi) 0.041 2399 0.108 3321	-0.02 -0.02
100 CD #13(6" from	center edge) 0.787 Strain Stress(psi) 0.039 3767 0.089 7535	-0.02 -0.02

104 CD #14(6" from center edge) 0.787 Strain Stress(psi) 0.075 5651	-0.02
93 CD #16(42" from center edge) 0.787 Strain Stress(psi) 0.117 9419 0.167 12056 0.251 15070	-0.00 -0.03 -0.01
113 CD #17(78" from center edge) 0.787 Strain Stress(psi) 0.078 6091 0.156 11421	-0.02 -0.05
101 CD #18(78" from center edge) 0.787 Strain Stress(psi) 0.066 5114 0.111 7672 0.145 9499	-0.02 -0.02 -0.05
96 CD #19(84" from center edge) 0.787 Strain Stress(psi) 0.044 4059 0.089 7012	-0.04 -0.05
66 CD #20(84" from center edge) 0.787 Strain Stress(psi) 0.100 8854 0.186 14317	-0.09 -0.02
67 CD #22(48" from center edge) 0.787 Strain Stress(psi) 0.117 10518	-0.09
64 CD #24(12" from center edge) 0.787 Strain Stress(psi) 0.116 10549	-0.05
116 MD PP TDO 10' from exit 1.181 Strain Stress(psi) 0.106 994 0.223 1242 0.295 1267 0.351 1491	-0.00 0.059 -0.00 -0.01
117 45 PP North Piece 0.787 Strain Stress(psi) 0.116 2693 0.171 3270 0.232 3463 0.302 4232	-0.02 -0.01 -0.02 -0.04

118	cđ	PP TDO 10'	from exit Strain St 0.141 0.200	1.181 ress(psi) 2411 2893	0.019 0.015
119	45	PP North F	Piece Strain St 0.131 0.206	0.787 ress(psi) 1883 2825	-0.05 -0.02
121	MD	PP TDO 10'	from exit Strain St 0.004 0.008	1.181 ress(psi) 0 0	-0.05 -0.02
Mat.	: Mate	erial, Thic	ck.: Thickn	ess, Ell. A: E	llipse Area,
Stra	in: E	ngineering	Strain in	MD of Instron.	
Stre	ss: E	ngineering	Stress in	MD of Instron.	
Test	Mat.	Orient. Ma	at. Note	Thick. (mils)	E (PSI)
31	MD	PP South P	Piece	0.787	33683
35	MD	PP South F	Piece	0.787	26045
40	CD	PP South F	Piece	0.787	87521
42	CD	PP South F	Piece	0.787	82490
17	45	PP South F	Piece	0.787	159618
20	45	PP South B	Piece	0.787	137483
22	MD	PP North H	lece	0.787	152377
20	MD	PP North H	Piece	0.787	L19985
30 22		PP North P		0.787	52493
34		PP North E		0.787	38495
24	45	PP North F	Diece	0.787	175739
57	Δ MD		from Exit	0.866	57444
62	MD	PP TDO 0'	from Exit	0.866	68026
58	MD	PP TDO 0'	from Exit	0.866	124495
61	MD	PP TDO 0'	from Exit	0.866	97318
56	CD	PP TDO 0'	from Exit	0.866	198847
53	CD	PP TDO 0'	from Exit	0.866	177258
60	CD	PP TDO 0'	from Exit	0.866	116325
71	CD	PP TDO 10	from exit	1.181	38269
70	CD	PP TDO 10	from exit	1.181	47392
45	MD	PP TDO 10	from exit	1.181	64533
49	CD	PP TDO 10	irom exit	1.181	56580
4/ 55		PP TDU IU'	from exit	1 250	00321 72654
22 50	MD MD	רער החיד מס ומכ החיד מס	· from exit	1 250	/ 3034 67051
91	עזיי רי	יייי מעד אייי	from avit	1.259	84417
94	CD	יחר 20 PP 100 20	from exit	1.259	106632
90	MD	PP TDO 30	' from exit	1.377	61133
72	MD	PP TDO 30	from exit	1.377	72181
86	CD	PP TDO 30	from exit	1.377	66155

114	MD	<pre>#11(96" from center edge)</pre>	0.787	86828
97	MD	#12(102" from center edge)0.787	72455
100	CD	<pre>#13(6" from center edge)</pre>	0.787	135067
104	CD	<pre>#14(6" from center edge)</pre>	0.787	81040
102	CD	<pre>#15(42" from center edge)</pre>	0.787	148527
93	CD	<pre>#16(42" from center edge)</pre>	0.787	135067
113	CD	<pre>#17(78" from center edge)</pre>	0.787	98575
101	CD	<pre>#18(78" from center edge)</pre>	0.787	98230
96	CD	<pre>#19(84" from center edge)</pre>	0.787	139274
66	CD	#20(84" from center edge)	0.787	168834
68	CD	#21	0.787	114807
67	CD	#22(48" from center edge)	0.787	122860
63	CD	#23(12" from center edge)	0.787	120849
64	CD	#24(12" from center edge)	0.787	106070
116	MD	PP TDO 10' from exit	1.181	22279
117	45	PP North Piece	0.787	41580
119	45	PP North Piece	0.787	31101
120.	CD	PP TDO 10' from exit	1.181	25591

Mat.: Material, Thick.: Thickness, E: Young's Modulus.

Test	Mat.	Orient. Mat. Note Thick.	(mils) Relax. time	(Sec)
31	MD	PP South Piece	0.787	780.3
29 34	MD CD	PP North Piece	0.787	1085
90	MD	PP TDO 30' from exit	1.377	1181
72	MD	PP TDO 30' from exit	1.377	629.9
87	MD	PP TDO 40' from exit	1.456	1158
77	CD	PP TDO 40' from exit	1.456	784.0
80	MD	PP TDO 50' from exit	1.574	603.0
43	MD	PP TDO 60' from exit	1.771	834.5
65	MD	PP TDO 60' from exit	1.771	907.6
36	MD	PP TDO 70' from exit	1.968	719.2
69	CD	PP TDO 70' from exit	1.968	991.3
109	MD	<pre>#5(42" from center edge)</pre>	0.787	977.0
116	MD	PP TDO 10' from exit	1.181	595.1
117	45	PP North Piece	0.787	980.9
118	CD	PP TDO 10' from exit	1.181	317.5
119	45	PP North Piece	0.787	739.8
121	MD	PP TDO 10' from exit	1.181	13.65

Relax. time: Relaxation time.

Test	Mat.	Orient. Mat. Note	Thick. (mils) Wave	form. (PSI)
31	MD	PP South Piece	0.787	859
35 40	MD CD	PP South Piece PP South Piece	0.787 0.787	$\begin{array}{c}1020\\4464\end{array}$
42	CD	PP South Piece	0.787	5710
20	45 45	PP South Piece PP South Piece	0.787	2855 971

22	MD	PP North Piece	0.787	2428
26	MD	PP North Piece	0.787	2671
30	CD	PP North Piece	0.787	1059
32	CD	PP North Piece	0.787	1076
34	CD	PP North Piece	0.787	1700
23	45	PP North Piece	0.787	1774
57	MD	PP TDO 0' from Exit	0.866	3082
62	MD	PP TDO 0' from Exit	0.866	3596
58	MD	PP TDO 0' from Exit	0.866	4646
61	MD	PP TDO 0' from Exit	0.866	3868
56	CD	PP TDO 0' from Exit	0.866	8220
53	CD	PP TDO 0' from Exit	0.866	6576
60	CD	PP TDO 0' from Exit	0.866	7535
45	MD	PP TDO 10' from exit	1.181	1539
47	CD	PP TDO 10' from exit	1.181	1107
55	MD	PP TDO 20' from exit	1.259	2537
59	MD	PP TDO 20' from exit	1.259	2590
94	CD	PP TDO 20' from exit	1.259	3194
24	CD	PP TDO 50' from exit	1.574	1700
25	CD	PP TDO 50' from exit	1.574	728
33	MD	PP TDO 60' from exit	1.771	431
43	MD	PP TDO 60' from exit	1.771	1889
50	MD	PP TDO 60' from exit	1.771	1674
65	MD	PP TDO 60' from exit	1.771	22144
52	CD	PP TDO 60' from exit	1.771	2616
54	CD	PP TDO 60' from exit	1.771	2846
69	CD	PP TDO 70' from exit	1.968	2673
93	CD 🕴	<pre>#16(42" from center edge)</pre>	0.787	4897
67	CD i	\$22(48" from center edge)	0.787	4798
64	CD 🕴	\$24(12" from center edge)	0.787	6028
116	MD	PP TDO 10' from exit	1.181	372
117	45	PP North Piece	0.787	1539
118	CD	PP TDO 10' from exit	1.181	1929
119	45	PP North Piece	0.787	1130
120.	CD	PP TDO 10' from exit	1.181	1205
Wave	form	· Wave formation or Onset	t of Waves	
nuve	TOTIN.	• wave formation of onset	C OL HUYCD.	
Test	Mat.	Orient. Mat. Note Thick.	(mils) Wave form.	(PLI)
31	MD	PP South Piece	0.787	0.677
35	MD	PP South Piece	0.787	0.803
40	CD	PP South Piece	0.787	3.515
42	CD	PP South Piece	0.787	4.496
17	45	PP South Piece	0.787	2.248
20	45	PP South Piece	0.787	0.765
22	MD	PP North Piece	0.787	1.912
26	MD	PP North Piece	0.787	2.103
30	CD	PP North Piece	0.787	0.834
32	CD	PP North Piece	0.787	0.847
34	CD	PP North Piece	0.787	1.338

0.787

0.866

1.396

2.67

23

57

45

MD

PP North Piece

PP TDO 0' from Exit

62 58 55 64 55 94 45 59 42 53 65 54 93 74 67 89 112 120	MDPPMDPPMDPPCDPPCDPPCDPPMDPPCDPP	TDO 0' from Exit0.866TDO 10' from exit1.181TDO 10' from exit1.181TDO 20' from exit1.259TDO 20' from exit1.259TDO 20' from exit1.574TDO 50' from exit1.574TDO 50' from exit1.771TDO 60' from exit1.771TDO 10' from exit1.81North Piece0.787TDO 10' from exit1.181North Piece0.787TDO 10' from exit1.181North Piece0.787TDO 10' from exit1.181	3.115 4.024 3.350 7.12 5.696 6.526 1.817 1.307 3.196 3.263 4.024 2.677 1.147 0.765 3.346 2.966 39.23 4.636 5.043 5.262 3.856 3.777 4.746 0.440 1.211 2.278 0.89 1.424
Test	Mat. Or:	ient. Mat. Note Thick. (mi	ls) Elastic Rec.
35	MD PP Elastic Elastic Elastic	South Piece 0.787 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1) Minor Axis Recovery (Mu-M1/M1)	$ \begin{array}{rcl} 0.047 \\ -0.13 \\ 0 & 0.076 \end{array} $
40	CD PP Elastic Elastic Elastic	South Piece 0.787 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1) Minor Axis Recovery (Mu-M1/M1)	$ \begin{array}{rcl} 0.059 \\ -0.05 \\ 0 & 0.035 \end{array} $
42	CD PP Elastic Elastic Elastic	South Piece 0.787 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1) Minor Axis Recovery (Mu-M1/M1)	$ \begin{array}{c} 0.042 \\ -0.05 \\ 0.071 \end{array} $
21	45 PP Elastic Elastic Elastic	South Piece 0.787 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1) Minor Axis Recovery (Mu-M1/M1)) = -0.09) = 0.2
26	MD PP Elastic	North Piece 0.787 Neck Recovery (Nu-N1/N1)=	

	Elastic Elastic	Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.212 -0.2
29	MD PP Elastic Elastic Elastic	North Piece 0.787 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0 -0.09 0
34	CD PP Elastic Elastic Elastic	North Piece 0.787 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	-0.05
57	MD PP Elastic Elastic Elastic	TDO 0' from Exit 0.866 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.069 -0.11 0.071
62	MD PP Elastic Elastic Elastic	TDO 0' from Exit 0.866 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.073 -0.13 0
61	MD PP Elastic Elastic Elastic	TDO 0' from Exit 0.866 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.020
53	CD PP Elastic Elastic Elastic	TDO 0' from Exit 0.866 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.020
60	CD PP Elastic Elastic Elastic	TDO 0' from Exit 0.866 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.021 -0.05 0.107
55	MD PP Elastic Elastic Elastic	TDO 20' from exit 1.259 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.068 -0.05 0.071
59	MD PP Elastic Elastic Elastic	TDO 20' from exit 1.259 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.069 -0.08 0.071
91	CD PP Elastic Elastic Elastic	TDO 20' from exit 1.259 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.021 -0.02 0.035

90	MD PP Elastic Elastic Elastic	TDO 30' from exit 1.377 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.261 -0.13 0.130
72	MD PP Elastic Elastic Elastic	TDO 30' from exit 1.377 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.071
87	MD PP Elastic Elastic Elastic	TDO 40' from exit 1.456 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	-0.08 -0.14 0.272
80	MD PP Elastic Elastic Elastic	TDO 50' from exit 1.574 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.064 -0.1 -0.41
25	CD PP Elastic Elastic Elastic	TDO 50' from exit 1.574 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	-0.13 0.076
33	MD PP Elastic Elastic Elastic	TDO 60' from exit 1.771 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.095 -0.13 0.083
43	MD PP Elastic Elastic Elastic	TDO 60' from exit 1.771 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.069 -0.13 0.090
50	MD PP Elastic Elastic Elastic	TDO 60' from exit 1.771 Neck Recovery (Nu-Nl/Nl)= Major Axis Recovery (Mu-Ml/Ml)= Minor Axis Recovery (Mu-Ml/Ml)=	0.022 -0.17 0.076
52	CD PP Elastic Elastic Elastic	TDO 60' from exit 1.771 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0 -0.05 0.071
36	MD PP Elastic Elastic Elastic	TDO 70' from exit 1.968 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)= Minor Axis Recovery (Mu-M1/M1)=	0.025 -0.09 0.083
51	CD PP Elastic Elastic	TDO 70' from exit 1.968 Neck Recovery (Nu-N1/N1)= Major Axis Recovery (Mu-M1/M1)=	-0.04 -0.08

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	Elastic	Minor	Axis	Recovery	(Mu-Ml/Ml	.)= 0	.071
69	CD PP Elastic Elastic Elastic	TDO 70 Neck H Major Minor)' fro Recove Axis Axis	om exit ry (Nu-Nl Recovery Recovery	1.968 /N1)= (Mu-M1/M1 (Mu-M1/M1	0 .)= -0 .)= 0	•23
109	MD #5 Elastic Elastic Elastic	(42" fi Neck H Major Minor	com ce Recove Axis Axis	enter edge ery (Nu-N1 Recovery Recovery) 0.787 /N1)= (Mu-M1/M1 (Mu-M1/M1	0 .)= -0 .)= 0	.082 .13 .090
93	CD #16 Elastic Elastic Elastic	(42" fi Neck H Major Minor	com ce Recove Axis Axis	enter edge ery (Nu-N1 Recovery Recovery) 0.787 /N1)= (Mu-M1/M1 (Mu-M1/M1	0 .)= -0 .)= 0	.045 .08 .071
116	MD PP Elastic Elastic Elastic	TDO 1(Neck H Major Minor)' fro Recove Axis Axis	om exit ery (Nu-Nl Recovery Recovery	1.181 /N1)= (Mu-M1/M1 (Mu-M1/M1	0 _)= -0 _)= 0	.061 .14 .166
117	45 PP Elastic Elastic Elastic	North Neck H Major Minor	Piece Recove Axis Axis	ery (Nu-N] Recovery Recovery	0.787 /N1)= (Mu-M1/M1 (Mu-M1/M3	0 _)= -0 _)= 0	.1 .1 .166
118	CD PP Elastic Elastic Elastic	TDO 1(Neck H Major Minor)' fro Recove Axis Axis	om exit ery (Nu-Nl Recovery Recovery	l.181 /N1)= (Mu-M1/M] (Mu-M1/M]	0 L)= -0 L)= 0	.043 .08 .034
119	45 PP Elastic Elastic Elastic	North Neck I Major Minor	Piece Recove Axis Axis	ery (Nu-N] Recovery Recovery	0.787 /Nl)= (Mu-Ml/M] (Mu-Ml/M]	0 L)= -0 L)= 0	.073 .05 .076
121	MD PP Elastic Elastic Elastic	TDO 10 Neck H Major Minor)' fro Recove Axis Axis	om exit ery (Nu-N] Recovery Recovery	l.181 /N1)= (Mu-M1/M] (Mu-M1/M]	-0 L)= -0 L)= 0	.03 .07 .076
Elast (Nu-N	ic Rec. N1/N1):	: Elas (Unload Neck d: dimens:	tic Re d Neck imensi ion.	ecovery. dimensic ion)/last	on-last me measured	easured Loaded	Loaded Neck
(Mu-M	41/Ml):	(Unload Axis d dimens	d Axis imensi ion.	s dimensio ion)/last	on-last me measured	easured Loaded	Loaded Axis

Test	Mat.	Orient.	Mat. 1	Note Thic	k. (mils) Ell.	Strains
31	MD	PP South	Piece Stra: 0.189 0.27 0.48	0 in Stress 9 1530 7 1748 7 2137	.787 (psi)	Major 0.117 0.171 0.318	Minor -0.13 -0.28 -0.37
35	MD	PP South	Piece Stra: 0.13 0.22	0 in Stress 0 1263 6 1505	.787 (psi)	Major 0.060 0.171	Minor -0.13 -0.20
40	CD	PP South	Piece Stra: 0.110	0 in Stress 0 9922	.787 (psi)	Major 0.060	Minor -0.13
42	CD	PP South	Piece Stra: 0.10	0 in Stress 3 8565	.787 (psi)	Major 0.060	Minor -0.13
26	MD	PP North	Piece Stra 0.11 0.20	0 in Stress 1 5586 9 7043	.787 (psi)	Major 0.060 0.030	Minor -0.06 -0.06
34	CD	PP North	Piece Stra: 0.12 0.21	0 in Stress 7 3400 7 4906	.787 (psi)	Major 0.060 0.117	Minor -0.13 -0.16
62	MD	PP TDO 0	from E: Stra 0.14 0.27	xit 0 in Stress 3 4795 5 6165	.866 (psi)	Major 0.060 0.171	Minor -0.13 -0.13
60	CD	PP TDO O	from E: Stra 0.14	xit 0 in Stress 3 1370	.866 (psi) 0	Major 0.060	Minor -0.13
59	MD	PP TDO 2	0' from Stra 0.15	exit l in Stress 1 3532	.259 (psi)	Major 0.117	Minor -0.13
91	CD	PP TDO 2	0' from 0 Stra 0.06 0.11	exit l. in Stress 4 4003 1 6357	259 (psi)	Major 0.030 0.060	Minor -0.06 -0.13
94	CD	PP TDO 2	0' from Stra 0.08	exit l. in Stress 4 4914	259 (psi)	Major 0.060	Minor -0.13
90	MD	PP TDO 30'	from exit 1.377 Strain Stress(psi) 0.117 2353 0.184 2660 0.239 2967 0.306 3171 0.390 3478 0.471 3785	Major 0.060 0.117 0.271 0.318 0.318 0.318	Minor -0.13 -0.16 -0.20 -0.24 -0.28 -0.33		
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72	MD	PP TDO 30'	from exit 1.377 Strain Stress(psi) 0.096 2484 0.202 2981 0.334 3577	Major 0 0.117 0.223	Minor -0.13 -0.20 -0.28		
85	MD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.089 2018 0.167 2337 0.217 2762	Major 0.030 0.089 0.117	Minor -0.09 -0.13 -0.20		
87	MD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.090 1995 0.196 2493 0.304 2793 0.402 3192	Major 0.030 0.089 0.171 0.271	Minor -0.13 -0.20 -0.28 -0.37		
83	CD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.103 4679	Major 0.030	Minor -0.06		
77	CD	PP TDO 40'	from exit 1.456 Strain Stress(psi) 0.106 5950 0.156 7969	Major 0.030 0.060	Minor -0.13 -0.13		
80	MD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.100 2166 0.248 3014 0.384 3579	Major 0.030 0.223 0.223	Minor -0.13 -0.28 -0.28		
79	MD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.114 2260	Major 0.030	Minor -0.13		
25	CD	PP TDO 50'	from exit 1.574 Strain Stress(psi) 0.193 10201 0.251 11901	Major 0.117 0.171	Minor -0.13 -0.20		
33	MD	PP TDO 60'	from exit 1.771				

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			Strain Stress(psi) 0.154 982 0.330 1360 0.450 1586	Major 0.117 0.223 0.318	Minor -0.13 -0.37 -0.28
43	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.111 2536 0.217 3346 0.353 4102 0.456 4318 0.512 4911 0.585 5289	Major 0.060 0.117 0.223 0.271 0.271 0.362	Minor -0.28 -0.20 -0.28 -0.28 -0.28 -0.37
50	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.132 3935 0.307 5525	Major 0.060 0.223	Minor -0.13 -0.20
65	MD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.135 34447 0.235 44289 0.412 57412	Major 0.060 0.117 0.223	Minor -0.13 -0.20 -0.28
52	CD	PP TDO 60'	from exit 1.771 Strain Stress(psi) 0.166 7009	Major 0.060	Minor -0.13
69	CD	PP TDO 70'	from exit 1.968 Strain Stress(psi) 0.103 3774 0.191 5346 0.279 6447 0.359 7233 0.432 7862	Major 0 0.117 0.171 0.318 0.485	Minor -0.13 -0.13 -0.20 -0.28 -0.28
105	MD	#l(6" from	center edge) 0.787 Strain Stress(psi) 0.111 3045	Major 0.060	Minor -0.09
103	MD	#2(12" from	center edge) 0.787 Strain Stress(psi) 0.055 3014 0.114 3579 0.206 4144	Major 0.030 0.060 0.117	Minor -0.06 -0.13 -0.20
106	MD	#4(30" from	center edge) 0.787 Strain Stress(psi) 0.055 3045 0.122 3807	Major 0.030 0.060	Minor -0.06 -0.13
109	MD	#5(42" from	center edge) 0.787		

				Strain 0.066 0.122 0.178 0.228 0.301 0.357 0.418 0.490	Stress(psi 3014 3579 4144 4521 4897 5274 5651 6028) Major 0.030 0.060 0.117 0.171 0.247 0.271 0.318	Minor -0.09 -0.13 -0.16 -0.20 -0.24 -0.28 -0.47 -0.37
108	MD	#6(48 "	from	center e Strain 0.066 0.106 0.172	edge) 0.787 Stress(psi 2983 3355 4101) Major 0 0.030 0.089	Minor -0.03 -0.06 -0.13
115	MD	#10(48"	from	center e Strain 0.061	edge) 0.787 Stress(psi 2583) Major 0.030	Minor -0.03
114	MD	#11(96 "	from	center e Strain 0.061 0.150 0.251 0.256	edge) 0.787 Stress(psi 2825 3767 4521 4521) Major 0.030 0.089 0.145 0.171	Minor -0.06 -0.09 -0.13 -0.16
97	MD ‡	#12(102 "	form	center e Strain 0.041 0.108	edge) 0.787 Stress(psi 2399 3321) Major 0 0.030	Minor -0.03 -0.06
100	CD	#13(6"	from	center e Strain 0.039 0.089	edge) 0.787 Stress(psi 3767 7535) Major 0 0.030	Minor -0.03 -0.06
104	CD	#14(6"	from	center e Strain 0.075	edge) 0.787 Stress(psi 5651) Major 0.030	Minor -0.06
93	CD	#16(42"	from	center e Strain 0.117 0.167 0.251	edge) 0.787 Stress(psi 9419 12056 15070) Major 0.060 0.089 0.117	Minor -0.06 -0.13 -0.13
113	CD	#17(78 "	from	center e Strain 0.078 0.156	edge) 0.787 Stress(psi 6091 11421) Major 0.030 0.060	Minor -0.06 -0.13

101 CD #18(78" from center edge) 0.787

			Strain 0.066 0.111 0.145	Stress(psi) 5114 7672 9499	Major 0.030 0.060 0.060	Minor -0.06 -0.09 -0.13
96	CD	#19(84" from	center e Strain 0.044 0.089	dge) 0.787 Stress(psi) 4059 7012	Major 0 0.030	Minor -0.06 -0.09
66	CD	#20(84" from	center e Strain 0.100 0.186	dge) 0.787 Stress(psi) 8854 14317	Major 0 0.171	Minor -0.13 -0.20
68	CD	#21	Strain 0.064	0.787 Stress(psi) 6405	Major O	Minor -0.06
67	CD	#22(48" from	center e Strain 0.117	edge) 0.787 Stress(psi) 10518	Major O	Minor -0.13
64	CD	#24(12" from	center e Strain 0.116	edge) 0.787 Stress(psi) 10549	Major 0.060	Minor -0.13
116	MD	PP TDO 10'	from exi Strain 0.106 0.223 0.295 0.351	t 1.181 Stress(psi) 994 1242 1267 1491	Major 0.060 0.171 0.197 0.271	Minor -0.06 -0.09 -0.20 -0.28
117	45	PP North P	iece Strain 0.116 0.171 0.232 0.302	0.787 Stress(psi) 2693 3270 3463 4232	Major 0.060 0.117 0.171 0.223	Minor -0.09 -0.13 -0.20 -0.28
118	CD	PP TDO 10'	from exi Strain 0.141 0.200	t 1.181 Stress(psi) 2411 2893	Major 0.089 0.117	Minor -0.06 -0.09
119	45	PP North P	iece Strain 0.131 0.206	0.787 Stress(psi) 1883 2825	Major 0.060 0.171	Minor -0.13 -0.20
121	MD	PP TDO 10'	from exi Strain	t 1.181 Stress(psi)	Major	Minor

	0.004	0		0.171	-0.20
Ell. strains:	True Strains Ellipses.	of Major	and Minor	axes of	the

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APPENDIX D

CALIBRATION FOR INSTRON SYSTEM

5 pound standard weight caused the following displacements: at 25 millivolts per centimeter: 5/8 inches; at 0.05 Volts per centimeter: 9/32 inches; at 0.25 Volts per centimeter: 1/16 inches; at 0.5 Volts per centimeter: 1/32 inches.

10 pound standard weight caused the following displacements: at 25 millivolts per centimeter: 1 1/8 inches; at 0.05 Volts per centimeter: 9/16 inches; at 0.25 Volts per centimeter: 1/8 inches;

Time ellapsed on the plotter was so close to that of the standard that the difference was negligible. The standard is 10 second = 10 second and 120 seconds = 120 seconds.

APPENDIX E

TEST BLANK

WEB HANDLING MATERIAL TESTING.

DATE/TEST NUMBER ______ Adhesive: Material: Thickness: in. (____mm) Test Specimen Grip: Roll ____Clamp Material Orientation: MD ____CD ___Other ____ # of times wrapped around bar: Test temperature: ____F (___C) Crosshead Speed: _____(in/min) Superimposed Grid ____Circles dia. ____in. ____Other description:

Observations:

Necking:

Center buckling:

Cross web waves:

End Failure:

Edge Failure:

Other:

VITA

Scott E. Robertson

Candidate for the Degree of

Master of Science

Thesis: DETERMINATION OF VARIATIONS IN CERTAIN MECHANICAL PROPERTIES ACROSS THE WIDTH OF A FINISHED PRODUCT OF POLYPROPYLENE WEB

Major Field: Mechanical Engineering

Biographical:

- Personal Data: Born in Tulsa, Oklahoma, November 22, 1963, the son of Dick Ross and Vida Marie Robertson.
- Education: Graduated from Broken Arrow High School, Broken Arrow, Oklahoma, in May 1982; recieved Bachelor of Science Degree in Mechanical Engineering from Oklahoma State University at Stillwater in May, 1986; completed requirements for Masters of Science degree at Oklahoma State University in May, 1989.
- Professional Experience: Teaching Assistant, Department of Mechanical Engineering, Oklahoma State University, August, 1986, to May, 1987. Web Handling Researcher, Department of Mechanical Engineering, Oklahoma State University, June, 1988, to January, 1989.